

SYNTHESIS OF THE NANOSTRUCTURED YAP:Ce VIA SPRAY PYROLYSIS BY POLYMERIC PRECURSOR SOLUTION

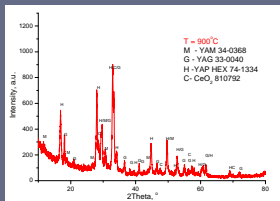
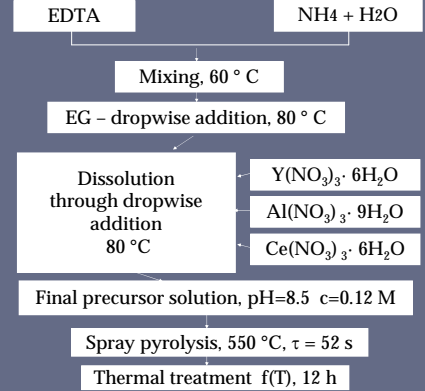
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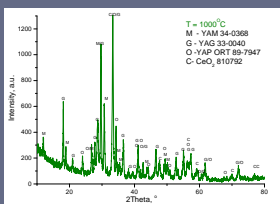
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Abstract The yttrium aluminum system ($Y_2O_3-Al_2O_3$) includes three compounds: yttrium aluminum garnet ($Y_3Al_5O_{12}$, YAG), yttrium aluminum perovskite ($YAlO_3$, YAP) and yttrium aluminum monoclinic ($Y_4Al_2O_9$, YAM). Doped with Ce YAP and YAG phases are well known optical materials used as a fast scintillators for synchrotron X-ray experiments. Synthesizing single YAP phase is difficult even through wet chemical processing because of the possible allocations of other phases. Here, we tried to synthesize fine powders of $YAlO_3:Ce^{3+}$ (5 at%) *via* spray pyrolysis of polymeric precursor obtained by dissolving the corresponding nitrates in ethylenediaminetetraacetic acid (EDTA) and ethylene glycol (EG) solution. Aerosol droplets are decomposed at 550 °C in argon atmosphere. In order to get a pure YAP:Ce phase as-prepared particles were additionally thermally treated in the range from 900 °C to 1100 °C for 12 hours in the air atmosphere.

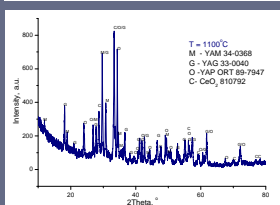
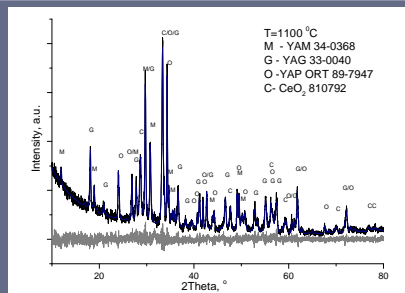
Schematic diagram for the synthesis of YAP powder



XRD analyses imply changeable phase compositions in thermally treated powders. The presence of a hexagonal YAP phase (*P63/mmc*) is observed at 900°C in ~70 wt%, while its orthorhombic modification is obtained at higher temperatures. With the rise of temperature, the content of the YAG phase increases from ~10 to 50 wt%. In all samples YAM and cerianite phase are also present as minority ones.

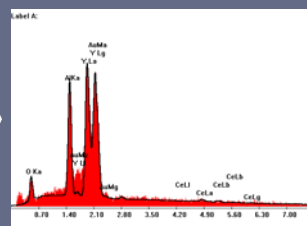
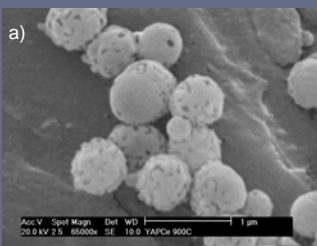


Example of the structural refinement is presented for the powder thermally treated at 1100°C.

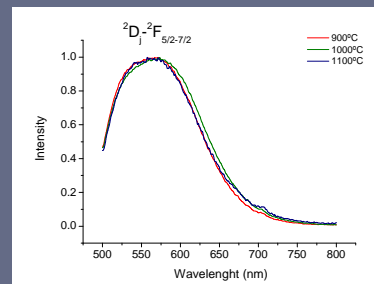
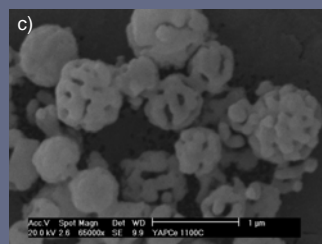
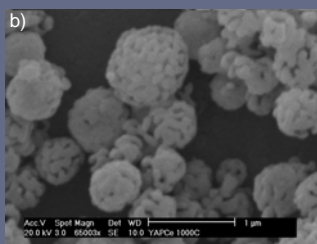


Microstructural parameters of determined phases (obtained through Rietveld refinement using Topas Academic software)

	900 °C, Gof = 1.59	1000 °C, Gof = 1.14	1100 °C, Gof = 1.12
YAP hexagonal [74-1334] S.G. P63/mmc	C.S. (nm) = 35(4) a(Å) = 3.6(7) c(Å) = 10.5(0)		
YAP orthorhombic [89-7947] S.G. Pbnm		C.S. (nm) = 269(0) a(Å) = 5.1(8) b(Å) = 5.3(2) c(Å) = 7.3(7)	C.S. (nm) = 215(1) a(Å) = 5.1(8) b(Å) = 5.3(2) c(Å) = 7.3(7)
YAG cubic [33-004] S.G. Ia-3d	C.S. (nm) = 56(0) a(Å) = 12.0(3)	C.S. (nm) = 84(9) a(Å) = 12.0(4)	C.S. (nm) = 84(9) a(Å) = 12.0(3)
YAM Monoclinic [34-0368] S.G. P21/a	C.S. (nm) = 19(7) a(Å) = 7.3(5) b(Å) = 10.5(9) c(Å) = 11.0(4)	C.S. (nm) = 76(4) a(Å) = 7.3(8) b(Å) = 10.4(5) c(Å) = 11.1(5)	C.S. (nm) = 72(6) a(Å) = 7.3(7) b(Å) = 10.4(4) c(Å) = 11.1(3)
Cerianite cubic [81-0792] S.G. Fm-3m	C.S. (nm) = 18(1) a(Å) = 5.49(9)	C.S. (nm) = 18(9) a(Å) = 5.4(0)	C.S. (nm) = 29(1) a(Å) = 5.4(0)



Highly spherical particle morphology is revealed with the SEM analyses. Particles are agglomeration-free and have high porosity. Volume precipitation is predominant. Temperature increase leads to the separation of the primary particles. Size of the secondary particles ranged from 200-800 nm; the mean particle size is around 350 nm. EDS analysis confirms required cations ratio (Y:Al ~1:1).



Photoluminescence emission spectra indicates wide green-yellow emission band with the maximum at 570 nm. This feature can be tentatively ascribed to the ${}^2D_1-{}^2F_{5/2-7/2}$ electron transition of Ce^{3+} ions in the YAG matrix.

SEM/EDS analyses of YAP:Ce³⁺ powders obtained *via* spray pyrolysis – thermally treated at: 900 (a), 1000 (b) and 1100 °C (c)

Conclusion Highly spherical, submicronic in size and agglomerated-free particles were obtained *via* spray pyrolysis method using polymeric precursor solution. Independently of the additional thermal treatment applied multiphase composition is confirmed in all samples. Target hexagonal YAP phase (70 wt%) is observed after thermal treatment at 900°C (12h), while additional rise of the temperature stabilizes the YAG phase generation and Ce³⁺ accommodation in it. Further synthesis optimization steps will be performed towards stabilization of kinetically favored pure YAP phase formation.

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